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# Green Synthesis of Bi<sup>3+</sup>-Mg<sup>2+</sup> Layered Doubled Hydroxides - MnO<sub>2</sub> Nanocomposites Supercapacitors

# P. Vanitha<sup>a</sup>, K. Karthikeyan<sup>b,\*</sup> and A. Thirumoorthi<sup>c</sup>

<sup>a</sup> Student, M. Sc. Chemistry, P. G. Department of Chemistry, Government Arts College, Udumalpet – 642 126, Tamilnadu, India.

<sup>b</sup> Guest lecturer, P.G. Department of Chemistry, Government Arts College, Udumalpet – 642 126, Tamilnadu, India.

<sup>c</sup>Assistant Professor, P.G. Department of Chemistry, Government Arts College, Udumalpet – 642 126, Tamilnadu, India.

# ABSTRACT

The  $Bi^{3+}-Mg^{2+}$  intercalated  $MnO_2 LDH$  has been synthesized by approach using Citric acid chemical route and lemon peel extract green synthesis route. The reducing agents of citric acid and lemon peel extract effects on synthesis of  $MnO_2$  in layered host of LDH has been investigated. The characterizations were done using by IR, UV-visible, XRD and SEM spectral studies. The electrochemical property in capacitance behaviour found using electrolytes 2M KOH solution 108 F g<sup>-1</sup> at a scan rate of 10 mV s<sup>-1</sup> for green synthesized  $MnO_2$  intercalated  $Bi^{3+}-Mg^{2+}$  LDHs.

Keywords: LDHs, MnO2, Green synthesis, Capacitance

# 1. Introduction

The anionic clay type as Layered double hydroxides (LDHs), was denoted the general formula  $[M^{2+}_{1-x}M_x^{3+i}OH_2](A^n_{x/n})_3.mH_2O$  where  $M^{2+}$  was a divalent cation  $(Mg^{2+}, Zn^{2+}, Cu^{2+}, Ni^{2+}, Co^{2+}$  etc.),  $M^{3+}$  was a trivalent cation  $(Al^{3+}, Fe^{3+}, Cr^{3+}$  etc.),  $A^n$  was an interlayer anion  $(CO_3^{2-}, SO_4^{2-}, NO_3^{-}, Cl^{-}, oH^{-}, etc.)$ , and x was the ratio of divalent to trivalent cations [1, 2]. LDHs have broad attentions since of their large specific surface area, memory effect, high chemical stability, adsorption property [3]. The  $A^{n-}$  anions, in the interlayer of the LDHs host layers, can be exchanged with other ones or species by using an intercalation reaction, many pillared LDHs materials with different morphology and property can be prepared, and the obtained pillared LDHs materials have been used for supercapacitor or lithium ion secondary batteries [4]. In general, supercapacitors include electrical double-layer capacitors (EDLCs) and pseudocapacitors based on the working principles. The energies in EDLCs are stored in the electric double-layer via electrostatic accumulation of charges while pseudocapacitors mainly utilize reversible Faradaic reactions to keep the energies, providing higher mass and/or volumetric specific capacitance [5– 8]. Normally, metal oxides (such as  $CoO_x$ ,  $MnO_2$ , and  $Fe_2O_3$ ) [9 – 11] and conducting polymers (PPy, PANI. and PEDOT) are widely utilized pseudo capacitive electrode materials.

Manganese dioxide ( $MnO_2$ ) has attracted great interest because of the variety of their structures and unique properties [12]. They find potential applications such as catalysts, solar cells and electrochemical capacitors, electrode materials of Li-ion and Li-air batteries [13-17] due to their low cost, earth abundance, environmental friendliness and variable oxidation states [18]. Liu et al. reported such a battery, the capacity of which was 285 mAh·g<sup>-1</sup> (MnO2), with capacity retention of 92% over 5000 cycles [19]. MnO2 is thus a promising candidate to be the electrode of next generation of commercial

\* Corresponding author. Tel.: 9360644240;

E-mail address: karthikathiravan2@gmail.com

rechargeable batteries. The next problem is thus to optimize the synthesis of this material in terms of cost and environmental friendliness.

Hongjuan Li et al. reported that  $Ni^{2+}$ - $Fe^{3+}$  layered double hydroxides (LDHs)/MnO<sub>2</sub> layered nanocomposite has been fabricated by using both layer-by-layer self-assembly method,  $Ni^{2+}$ - $Fe^{3+}$  LDHs/MnO<sub>2</sub> nanocomposite exhibits a relative good capacitive behavior in a neutral electrolyte system, and its initial capacitance value is 104 F g<sup>-1</sup>[20]. Lianlian Liu et al. synthesized the MnO<sub>2</sub> nano sheets were directly deposited on the NiCo-LDH to form three-dimensional (3D) self-supported core-shell MnO<sub>2</sub>/NiCo-LDH/CC flexible electrodes and found capacitive performance of 312 F g<sup>-1</sup>[21].

Qing Sun et al. [22] found NiFe-LDH@FeOOH nanocomposites present high specific capacitance (1195 F/g at a current density of 1 A/g), lower resistance and well cycling performance (90.36% retention after 1000 cycles). Besides, the NiFe-LDH@MnO<sub>2</sub>//NiFe-LDH@FeOOH supercapacitor exhibits 22.68 Wh/kg energy density at 750 W/kg power density, demonstrating potential application in energy storage devices. Hao Luo et al. found Core-sheath hierarchical-architectured materials MnO2@Co-Ni LDH possesses highly micro-structural integrity and using as positive electrode for pseudocapacitor, this material exhibits large specific capacitance of 1436 F  $g^{-1}$  at a current density of 1 A  $g^{-1}$ .

Bismuth based materials [23] possess exclusive band structures and high photo-corrosion stability. The pervoskite BiFeO<sub>3</sub> nanocrystalline thin film electrode exposed comparable specific capacitance of 81 Fg<sup>-1</sup> and electrochemical supercapacitive performance and stability in an aqueous NaOH electrolyte to that of commonly used ruthenium based pervoskites. The pervoskite BiFeO3 nanocrystalline electrode used for supercapacitor applications [24, 25].

In this study, we report  $MnO_2$  intercalated  $Bi^{3+}-Mg^{2+}$  LDHs nanocomposites were synthesized by two methods one is chemical route using citric acid and another one is eco-friendly route lemon peel extract used as a reducing agent. The chosen of lemon peel since it has three reducing reagents flavanoid glycoside, p-coumaric acid and  $\beta$ - sitosterol. The effect of reducing agents on synthesize of  $Bi^{3+}-Mg^{2+} - MnO_2$  LDHs nanocomposites and investigated their behavior of optical, crystalline size, morphology and capacitance. The as-prepared samples were characterized by FT-IR, UV, XRD, SEM, and Photoluminescence. The electrochemical properties of both samples were evaluated by cyclic voltammetry.

# 2 EXPERIMENTAL SECTION

#### 2.1 Lemon peel extracts preparation

10 g of peel of lemon were successively washed, dried at 90 °C, and boiled in distilled water for 10 min at 100° C. The peel extract was filtered.

# 2.2 Preparation of Bi<sup>3+</sup>– Mg<sup>2+</sup> LDHs

Equimolar ratio of Bi(NO<sub>3</sub>)<sub>3</sub> .9H<sub>2</sub>O and Mg(NO<sub>3</sub>)<sub>2</sub> .6H<sub>2</sub>O in a molar ratio dissolved in 100ml deionized water and added 1 M HNO<sub>3</sub> (2ml) stirred well for homogeneous mixture. The solution of metal nitrates was slowly poured into the solution NaOH solution maintained pH – 12 under vigorous stirring. After mixing, the obtained slurry was aged at 80°C for 6 h, filtered, washed with distilled water and dried in an oven at 60 °C.  $Bi^{3+} - Mg^{2+}$  LDHs [BMLDHs] was attained [26].

#### 2.3 Synthesize MnO<sub>2</sub> intercalated LDHs nanocomposites

LDHs material (2 g) was treated with 1.2 mol  $L^{-1}$  aqueous solution of KMnO<sub>4</sub> for 1 day at room temperature MnO<sub>4</sub> – LDHs material was obtained. The precipitate was isolated by filtration and washed several times with distilled water to remove potassium ions and dried 60 °C. MnO<sub>4</sub>–LDHs obtained.

Then the  $MnO_4$ -LDHs added with 50ml distilled water and then add 50ml peel extract after stirring vigorously for 5 h at room temperature this mixture changed in color from pale brown to dark brown due to reduction finally  $MnO_2$  – LDHs synthesized. The collected precipitate was dried overnight at 60 °C [27]. The sample has been given description as BMLDHM– LP.

In chemical route,  $MnO_2 - LDHs$  synthesized same manner described above however citric acid has been added as an alternative of lemon peel extract. The sample has been given description as BMLDHM- C.

# **3 RESULTS AND DISCUSSION**

#### 3.1 UV – VISIBLE

Hongjuan Li et al. found that multilayer growth of  $(Ni^{2+}-Fe^{3+}LDHs/MnO_2)_n$  is examined by recording the UV-vis absorption spectral studies. A broad absorption band centered at around 365 nm is characteristic of manganese oxide nanosheets. While the  $Ni^{2+}-Fe^{3+}$  LDHs nanosheets deposited on quartz slide show no optical absorption in the wavelength range of 200–800 nm. The absorbance at 365 nm increases almost nearly linear in proportion to the number of layer pairs, indicating the  $Ni^{2+}-Fe^{3+}$  LDHs nanosheets and MnO<sub>2</sub> nanosheets are successfully assembled [20].

In Fig. 1 shows UV- visible spectral of LDHM – C and LDHM – LP nanocomposites respectively. The peak found the region at 310 - 325 nm due to MnO<sub>2</sub> present in the layered host of LDHs. The absorbance LDHM – LP has been increased due to MnO<sub>2</sub> assembled well in the layered space of LDHs. Meanwhile, Bi<sup>3+</sup>–Mg<sup>2+</sup> LDHs has no optical absorption.



Fig. 1 – UV-visible spectra of synthesized nanocomposites (a) BMLDHM-C; (b) BMLDHM-LP.

# Tauc's plot

#### Energy band of materials is related to absorption coefficient *α* by the Tauc's relation:

ahv = A(hv - Eg)n

Where A is a constant, hv the photon energy, Eg the band gap and n is an index which assumes values 1/2, 3/2, 2 or 3 depending on the nature of the electronic transition responsible for the absorption. n=1/2 is taken for an allowed direct transition. Therefore, by plotting a graph between  $(\alpha hv)^2$  and hv in eV, a straight line is obtained which gives the value of the direct band gap. The extrapolation of straight line to  $(\alpha hv)^2 = 0$  gives the value of the direct band gap of the material. Fig. 2 (a) and (b) shows that tauc'polt and found band gap 3.67 eV and 3.75 eV for synthesized BMLDHM-C and BMLDHM-LP nanocomposites respectively.



Fig. 2 – Tauc's plot of synthesized nanocomposites (a) BMLDHM-C; (b) BMLDHM-LP.

# **3.2 XRD ANALYSIS**

Zong – Huai Liu et al. suggested that  $MnO_2$  pillared  $Ni^{2+}$ –  $Fe^{3+}$  layered double hydroxides nanocomposite has been successfully fabricated using an intercalation/reduction reaction followed by heating treatment. The precursor,  $Ni^{2+}$ –  $Fe^{3+}$  LDHs has a basal spacing of 0.78 nm and no peaks of impurities are discerned, indicating a high purity of the product. After intercalation of  $MnO_2$  particles, the interlayer basal space increased to be 0.835 nm [28].

In Fig. 3 exposed that XRD pattern of layered doubled hydroxides with LDHM – C and LDHM – LP nanocomposites. The peaks at  $2\Theta = 11^{\circ}$  and  $22^{\circ}$  corresponding to the plane 003 and 006 respectively. The basal space of 003 plane of pure BMLDHs has been found d=0.785 further which was increased since intercalation of MnO<sub>2</sub>. That are confirmed by basal spaces of samples LDHM – C and LDHM – LP those are found d=0.84 and 0.88 respectively. Hence XRD pattern confirms the MnO<sub>2</sub> intercalated in the layer space of LDHs. The diffraction peaks assigned that rhombohedral structure of LDH meanwhile MnO<sub>2</sub> intercalated LDHs structure has been compared with pristine Co<sup>2+</sup> – Al<sup>3+</sup> LDH (CALDH) [29]. Moreover, the basal reflection of BMLDH-LP expanded, indicating the layer structure of the samples increased due to intercalation of MnO<sub>2</sub>.



Fig. 3 - XRD spectra of synthesized nanocomposites (a) BMLDHMs ; (b) BMLDHM-C ; (c) BMLDHM-LP.

Table 1 - Indexing of XRD patterns of synthesized nanocomposites.

Sample	d <sub>003</sub> (nm)	d <sub>006</sub> (nm)	d <sub>009</sub> (nm)	d <sub>110</sub> (nm)	Crystallite size in 'c' direction (nm)	Crystallite size in 'a' direction (nm)
CALDH	0.75264	0.37691	0.25126	0.15363	2.26023	0.30726
BMLDHs	0.73081	0.38225	0.24851	0.15077	2.24084	0.30154
BMLDH – C	0.83050	0.38616	0.24794	0.15037	2.41745	0.30074
BMLDH – LP	0.90133	0.39733	0.28123	0.15123	2.51886	0.30246

# 3.3 SEM analysis

Zhe Yan et al [30]. reported that  $MnO_2 - pillared Co^{2+} - Ni^{2+} - Fe^{3+}$  layered double hydroxide nanocomposite with porous structure has been successfully prepared by using an intercalation/reduction reaction and followed by heating treatment at 200°C. The precursor, LDHs material, is consisted of thin hexagonal platelets with a mean lateral size of about 300 nm.  $MnO_2$ -LDHs material hexagonal platelet morphology was observed. Min Lia et al [31]. suggested that the flower – like Ni – Fe layered double hydroxides loaded on Ni foam (NF/NiFe LDHs) are synthesized via one-pot hydrothermal method. The SEM images of the sample picked at different growth time nanowire and flower like shape observed.



Fig. 4 - SEM morphology of synthesized nanocomposites (a) BMLDHMs ; (b) BMLDHM-C; (c) BMLDHM-LP.

The Fig.4 shows that SEM images (a), (b) and (c) of BMLDHs, LDHM – C and LDHM – LP respectively. In Fig. 4 (a) shows SEM images of BMLDHs which observed platelet morphology mean while Fig. 5(b) LDHM – C exposed that irregular shape since intricate the growth of the  $MnO_2$  nanoparticles formation during reduction, which explains the smaller size of the primary particles and size found to be 53nm. As shown in Fig. 4(c) LDHM-LP consists of fairly growth of nanorods and spherical shape of morphology observed an average size about 68 nm. It is clearly demonstrated that the nature of the reducing reagent in lemon peel has a strong effect on the morphology which support the growth of nanorods due to abundant phenolic compound p-coumarin acid, cholesterol similar structure  $\beta$ -sitosterol and flavonoid glycosides are anti-oxidative components in peel.

# 3.5 IR Spectral analysis

The broad band observed at 3458 cm<sup>-1</sup> was attributed to O - H stretching modes of interlayer water molecules and H - bonded OH groups, and the peak at 1633 cm<sup>-1</sup> corresponding to bending mode of water molecules. The bands in the range of 800 – 400 cm<sup>-1</sup> were due to the Metal – Oxygen and Oxygen – Metal – Oxygen groups of lattice vibration bands. The LDHM – C and LDHM – LP shows a peak at 573 cm<sup>-1</sup> indicates that the formation of the MnO<sub>2</sub> phase in the layered host. In Fig. 5 depicts the FT – IR spectra of BMLDHs, BMLDHM – C and BMLDHM – LP nanocomposites. It shows that peak of metal – oxygen bond 400 – 700 cm<sup>-1</sup> besides interlayer water molecules peak intensity at the range of 1630- 1640 cm<sup>-1</sup> was decreased since MnO<sub>2</sub> intercalated into layered host of BMLDHs hence the peak was raised at 526 – 538 corresponding to MnO<sub>2</sub>. The table 2 shows that the vibration peak position and assignment of the nanocomposites [32, 33].

![](_page_4_Figure_4.jpeg)

Fig. 5 – FT-IR spectra of synthesized nanocomposites (a) BMLDHMs; (b) BMLDHM-C; (c) BMLDHM-LP.

Table 2 - FT-IR peak position and assignments comparison of synthesized nanocomposites.

BMLDHs BMLDHM - C		BMLDHM – LP	Assignments	
427.25, 622.06 and 664.50	468.72 and 686.68	449.43 and 478.36	Metal – Oxygen bonds (Bi – O, O – Mg / Bi – O)	
_	526.58	538.16	MnO <sub>2</sub> Phase	
1630.87	1635.69	1639.55	Bending mode of H <sub>2</sub> O	
3431.48	3443.05	3427.62	Interlayer O – H stretching modes	

# **3. 4 CYCLIC VOLTAMMETRY**

F. Li et al. synthesized low-cost high-performance asymmetric supercapacitors based on  $Co_2AlO_4 - MnO_2$  nanosheets and  $Fe_3O_4$  nanoflakes and suggested electrochemical properties of these composites displays a typical pseudocapacitive behavior with obvious redox peak clearly observed which can be assigned to the reaction of the Ni<sup>2+</sup>/Ni<sup>3+</sup> and Fe<sup>2+</sup>/Fe<sup>3+</sup> associated with OH<sup>-</sup>. Besides, larger area of CV curves, showing much higher pseudocapacitance [34]. It derives from the extra pseudocapacitance from the contribution of the  $MnO_2$  shell, which incorporates K<sup>+</sup> cations on the outer surface and/or possible intercalation and deintercalation of K<sup>+</sup> ions.

![](_page_5_Figure_1.jpeg)

Fig. 6 - XRD spectra of synthesized nanocomposites (a) BMLDHM-C; (b) BMLDHM-LP.

The cyclic voltammograms (CVs) shown in Fig. 6. The discharge process broad current peak around 2.0 V. The cyclic voltammograms studies of LDHM – LP and LDHM – C have been investigated for their capacitance behavior in 2M KOH solution as the electrolyte at various scan rate  $10mVs^{-1}$  –  $50mVs^{-1}$ . The broadness of redox peaks is attributed to the poor crystallinity and morphology of MnO<sub>2</sub> in LDHM – C. The narrow peak was found at 1.5 eV for LDHM – LP since uniform growth MnO2 with good crystalline in nature. The capacitance found sample LDHM – LP and LDHM – C has  $108 \text{ Fg}^{-1}$ and  $62 \text{ Fg}^{-1}$  respectively. The pseudocapacitive behavior apparent redox peak observed which can be confirmed to the reaction of the Bi<sup>2+</sup>/Bi<sup>3+</sup> and Mg<sup>2+</sup>/Mg<sup>3+</sup> reacted with hydroxyl ion in electrolyte. Besides, the better electrochemical performance of LDHM-LP is attributed due to crystalline nature and fair growth of MnO<sub>2</sub> in the layered host of BMLDHs.

#### 4. Conclusions

The LDH intercalated  $MnO_2$  synthesized (BMLDHs) successfully by using chemical route – citric acid (BMLDHM – C) and green synthesis – lemon peel (BMLDHM – LP). Both the crystallization and the electrochemical properties are enhanced in the BMLDHM – LP with respect to BMLDHM – C. This is ascribed to the fact that the lemon peel contains three reducing which leads to better results in the BMLDHM – LP. The IR spectral studies confirms that  $MnO_2$  intercalated in the layered host of BMLDHs. UV –Visible spectra exposed that the intensity of peak of BMLDHM – LP is enhanced due to substantial intercalation. The XRD spectral studies show that the basal space of BMLDHs improved for BMLDHM – LP. SEM morphology found that lemon peel forms nanorods with spherical shape of crystalline  $MnO_2$  which improves electrochemical properties of the BMLDHM – LP are as fine as those obtained with chemical reducing reagents. Moreover, using lemon peel has diminished both costly and polluting. This new way of synthesis of  $MnO_2$  intercalated BMLDHs using for many applications in the industry, including electrochemical energy storage.

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