



Surface-Modified Nanoparticles Reinforced Polymer Nanocomposites

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ABSTRACT

In this work, poly(butylene terephthalate)-block-poly(tetramethylene glycol) (PBT-PTMG)-based thermoplastic elastomer (TPE)/nano titan dioxide (nTiO₂) nanocomposites were compounded by melt mixing process. Before compounding, the surface of the nanoparticles was modified with chloroacetic acid for better surface bond and good dispersion. The effect of unmodified and surface-modified nTiO₂ particles with a varying concentration on the mechanical and thermal features of TPE/nTiO₂ nanocomposites was made and evaluated. Samples of tensile and impact tests were obtained by a hot press machine. The outcomes showed that chloroacetic acid-containing nanocomposites gave a strong interfacial bond between nanoparticles and matrix due to chemical linkage while the absence of chloroacetic acid-induced agglomerates. The presence of chloroacetic acid increased tensile strength and impact strength but tensile modulus reduced due to its limited amount of usage in the coating of the nanoparticles. The inclusion of nTiO₂ in TPE significantly increases the overall thermal properties of the matrix such as the thermal degradation temperature and thermal stability of nanocomposites.

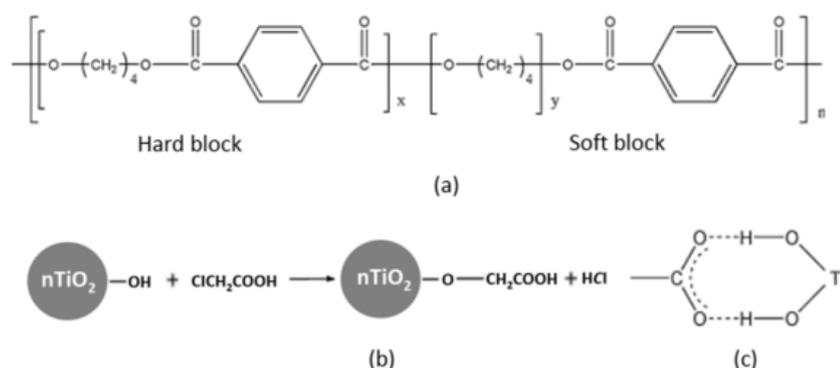
Keywords: Thermoplastic elastomer, nano-TiO₂, nanocomposites, mechanical properties, Thermal properties.

1. INTRODUCTION

Poly(butylene terephthalate)-block-poly(tetramethylene glycol) (PBT-PTMG)-based thermoplastic polyester elastomers (TPEs) are new ingredients that are gaining such importance across the boundaries between rubber and plastic. Thus, research on the viability of TPEs is important and required. Due to the phase-isolated microstructure, it has a unique combination of strength, flexibility, and processing. The TPE consisting of poly(butylene terephthalate) (PBT), as a hard portion, and polytetramethylene glycol (PTMG) is a soft portion, have superior mechanical, thermal properties, and elasticity (Jiang et al., 2019). It has attracted interest due to its low specific gravity and great reprocessing, which has been considered as an environmentally friendly material that replaces vulcanized rubber and PVC. TPEs are also used for gears, tubing, electronics, electronic parts, and even increasingly substitute the cross-linked rubbers (Yao and Yang, 2010). The basic structure of the PBT-PTMG copolymer-based TPE is revealed in Scheme 1(a). TPE shows outstanding mechanical properties (Nagai et al., 1997) at temperatures of 130°C combined with very good low-temperature flexibility. To improve the mechanical, thermal, and electrical properties of TPE, particulate fillers such as CaCO₃, SiO₂, ZnO, and TiO₂, etc. are added. Inorganic nano-metal oxide (nTiO₂) is one of them because, among all other semiconductor metal oxides families, nTiO₂ is inexpensive, electronic and optical stable, nontoxic, and corrosion-resistant. The introduction of nano-TiO₂ into polymers can improve the mechanical and thermal properties of polymers due to the small size of organic polymers and inorganic nanoparticles and the strong interfacial interaction between large specific fields and quantum effects. As a result, these nanocomposites can be widely applied in coatings, rubbers, plastics, sealants, fibers, and other applications (Li et al., 2002; Takai et al., 1998; Wu et al., 2000). In this study, we used TPE as a matrix and made TPE/nTiO₂ nanocomposites by the melt mixing process. It is known that nTiO₂ can react with acidic functionalities of polymers, increasing the particle-matrix interaction. The surface modification of nTiO₂ with chloroacetic acid and its hydrogen bonding are shown in Schemes 1 (b) and (c), respectively.

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Scheme 1- (a) Basic structure of thermoplastic polyester elastomer (TPE); (b) surface modification of TiO_2 with chloroacetic acid (c) binding mode of carboxylate with titania surface: hydrogen bonding.

TPE/ $n\text{TiO}_2$ nanocomposites continue to arouse great interest in scientists due to their flexibility in design in addition to advanced mechanical strength, good chemical, electrical and thermal properties (Rong et al., 2001). Despite a significant increase in mechanical properties, nanoparticles may agglomerate and thus reduce the properties of nanocomposites. To overcome this problem, dispersant and coupling agents are being used (Kruenete et al., 2004). In addition, the loss of impact strength due to the stiffness of inorganic ingredients is usually another problem. Researchers use ingredients of elastomeric nature to increase the toughness of composites (Bao and Tjong, 2007; Ishak et al., 2008). In this article, a simple technique for preparing novel TPE/ $n\text{TiO}_2$ nanocomposites are reported. A highly homogeneous dispersed CnTiO_2 /TPE composite was formed by interacting between nanofiller and matrix through covalent bonds and interacting with the acidic functionality of the polymer with $-\text{COOH}$ moieties on the surface of carboxylated $n\text{TiO}_2$ particles. In this study reported herein, the effect of the inclusion of uncoated and coated $n\text{TiO}_2$ particles and particle content on the mechanical and thermal properties of the particular TPE.

2. EXPERIMENTAL

2.1. Materials and sample preparation

The polymer matrix TPE granules were kindly delivered by M/S Rupal Plastics Ltd, Mumbai, India with a grade was Hytrel 6356. The $n\text{TiO}_2$ under the trade name of Aeroxide[®] P25 was supplied by Evonik Industry (Germany). Chloroacetic acid (AR) from Shanghai Chemicals Co., was used without further purification.

The $n\text{TiO}_2$ suspension (90-ml water, 5 g of $n\text{TiO}_2$) was stirred for 1 h and ultrasonicated for 20 min, then mixed with an ascertained amount of chloroacetic acid dissolved in 10 ml of deionized water. The mixture was heated to 80°C , and the reaction was performed for 2 h under air with continuous stirring. The pH value of the upper liquid layer was neutralized until the carboxylated $n\text{TiO}_2$ was purified by centrifugation and then washed with distilled water and no AgCl was detected when the AgNO_3 aqueous solution was added to the liquid. The quantity of carboxylated $n\text{TiO}_2$ was measured by acid-base titration. The $n\text{TiO}_2$ suspension (50 ml) was titrated with 0.005 M standard alkaline solution using phenolphthalein as an acid-base indicator.

Uncoated and coated $n\text{TiO}_2$ particles were added separately to the TPE in different ratios as 1, 3, and 5 wt%, and then extruded in a Berstorff twin-screw extruder (ZE-25A UTX, KraussMaffei Berstorff GmbH, Germany). The coding of the nanocomposites was done as follows: TPE/1Un TiO_2 , TPE/3Un TiO_2 , and TPE/5Un TiO_2 for uncoated and TPE/1Cn TiO_2 , TPE/3Cn TiO_2 , and TPE/5Cn TiO_2 for coated. The diameter of the screws was 25 mm and the length/diameter ratio (L/D) was 44. The processing temperature was ranged from 200°C neighboring the hopper to 220°C at the die and the speed of the screw was 100 rpm. Each TPE/ $n\text{TiO}_2$ composite was extruded three times to ensure better dispersion quality of the nanoparticles. The extrudate strands were granulated and dried at 80°C under vacuum for 3 h.

2.2 Characterizations

The tensile test was measured using Shimadzu Universal Testing Machine (model AG-1, Japan). Samples were ~ 10 mm wide, 1 mm thick, with a gauge length of 50 mm. A crosshead speed of 10 mm/min was used to measure the tensile strength and tensile modulus. The Izod impact test was determined on notched specimens using a 5.4 J pendulum hammer on a Zwick impact test machine. Five replicates of tensile measurement were performed for each formulation. Transmission electron microscopy (TEM, JEM-2100F, JEOL) was used to examine the dispersion of TPE and $n\text{TiO}_2$. Scanning electron microscopy (SEM) examination was performed with a JSM-6360LV scanning electron microscope, and the specimens were coated with a thin layer of gold before SEM examination. The crystallization and melting behaviors of nanocomposites were investigated by differential scanning calorimetry (DSC, Perkin Elmer DSC-7) in the N_2 atmosphere. Each sample (5-8 mg) was heated from 30 to 240°C at a heating rate of $10^\circ\text{C}/\text{min}$ and then held at 240°C to confirm duplicate thermal history. The sample was cooled down to 30°C at a cooling rate of $10^\circ\text{C}/\text{min}$. The thermal stability of the nanocomposites was determined using a thermogravimetric analyzer (TGA, TA Instruments Q500) with a heating rate of $20^\circ\text{C}/\text{min}$ in an N_2 atmosphere.

3. RESULTS AND DISCUSSION

3.1. Mechanical properties of the nanocomposites

Metal oxide nanoparticles progress the mechanical features of the polymer matrix (Rahman et al., 2019). Typically, the tensile strength and elastic modulus increase while the impact strength decreases. Variation of tensile strength according to TPE/uncoated nTiO₂ (TPE/UnTiO₂) and TPE/coated nTiO₂ (TPE/CnTiO₂) nanocomposites and nTiO₂ content is illustrated in Figure 1 (a). It was informed that the tensile features of nanocomposites depend on the interaction and characteristics between the composition content (Zaman et al., 2014). It can be observed that the tensile strength (TS) increased with increasing nTiO₂ content up to 3 wt% nTiO₂ loading (23% increment in tensile strength for 3% loading) and then decreased by increasing nTiO₂ loading to 5 wt% (5% decrement in tensile strength than 3% loading). The inclusion of nTiO₂ particles to the TPE improves the tensile strength of the matrix at 3 wt% nTiO₂ due to strong stress transfer from TPE to nTiO₂. At 5 wt% of nTiO₂ particles, an improvement in tensile strength was reduced due to the relative low-stress transfer effect of nanoparticles. These outcomes were compatible with the results of the research study performed by Zaman et al. (Zaman et al., 2013). To progress the filler dispersion progression in the TPE matrix and to stretch the interfacial bonding between nTiO₂ and TPE, chloroacetic acid is led to modify the TPE/nTiO₂ nanocomposites. The inclusion of TPE/CnTiO₂ nanocomposite has created significantly noticeable interfacial bonding between filler and matrix than those of TPE/UnTiO₂ nanocomposite. The maximum tensile strength of the TPE/CnTiO₂ nanocomposite was 24.9 MPa at 3 wt% of nTiO₂ content and about 23% higher than that of the TPE matrix. A highly homogeneous dispersed CnTiO₂/TPE nanocomposite was formed by interacting between nanofiller and matrix through covalent bonds and interacting with the acidic functionality of the polymer with -COOH moieties on the surface of carboxylated nTiO₂ particles.

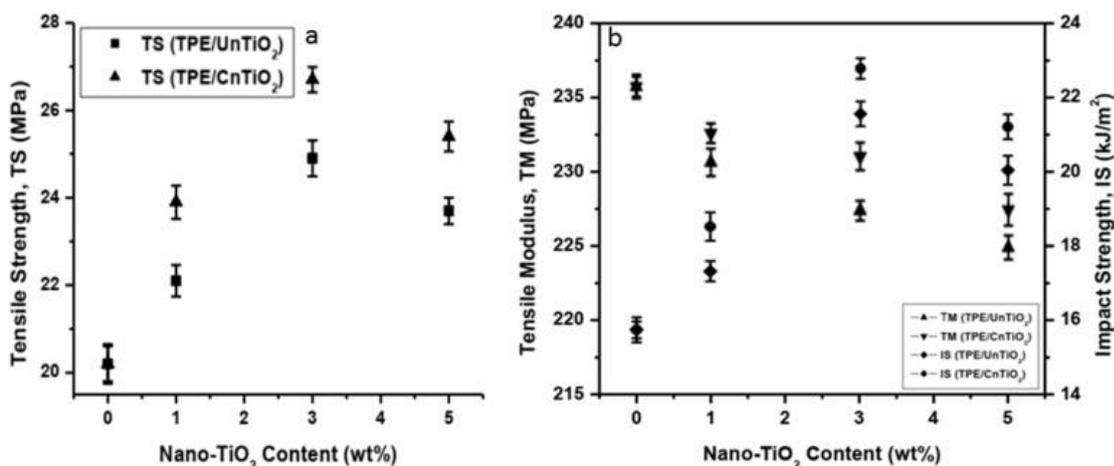


Fig. 1- (a) Tensile strength and (b) tensile modulus and impact strength of TPE/nTiO₂ nanocomposite.

Figure 1 (b) displays the variation of tensile modulus and impact strength of the nanocomposites against nTiO₂ content. It can be observed that impact strength (IS) increased with increasing nTiO₂ content up to 3 wt% nTiO₂ loading for uncoated and coated nanocomposites and then decreased, but tensile modulus (TM) steadily declined with increasing nTiO₂ content. Maximum IS (22.9 kJ/m²) was obtained at 3 wt% of CnTiO₂ content. At this stage, the IS value of the TPE/CnTiO₂ nanocomposite was about 18% higher than that of the TPE/3UnTiO₂ and even 46% higher than that of the TPE matrix. The presence of chloroacetic acid provided the toughness of the composites through the molecular flexibility of the polymer matrix and induced this higher IS but lower TM as seen in Figure 1 (b). The reason for lower TM than that of TPE was the plasticizing effect of the chloroacetic acid.

3.2. Surface morphology and particle dispersion

Mechanical properties are related to the uniformity of multiple morphologies, domain sizes, and shapes to determine the degree of dispersion of interactions between the two stages, and it is considered to be one of the most important elements (Ghazy et al., 2020). Figure 2 (a)-(b) represents the TEM microphotographs of TPE/3UnTiO₂ and TPE/3CnTiO₂ nanocomposites, respectively. Figure 2 (a) shows considerably bigger nTiO₂ particles, which are not intercalated and probably form a 'micro composite' structure, possibly due to the absence of surface carboxylated nTiO₂. The black shape exhibits the nTiO₂ tactoids and the rest of the region represents an uninterrupted TPE. Nevertheless, some black shapes may indicate some weakly dispersed nTiO₂ aggregates. Figure 2 (b) on the other hand reveals relatively small nTiO₂ particles compared to Figure 2 (a) and the nTiO₂ particles were separated into lighter parts by the blending manner. Therefore, surface carboxylated nTiO₂ should be mixed with TPE to better disperse nTiO₂ in the matrix and to increase the tensile features.

Figure 2 (c)-(d) illustrates the SEM microphotographs of TPE/3UnTiO₂ and TPE/3CnTiO₂ nanocomposites, respectively. The particles of nTiO₂ in Figure 2(c) were randomly distributed in the TPE matrix and some large portions were exposed above the fracture surface. Big elements are distributed in TPE to

TPE/3UnTiO₂ nanocomposite so that no functional polymer is present and the interfaces appear to be individually wet and/or weak to the adhesion of the components. As shown in Figure 2 (d), the TPE/3CnTiO₂ system was more uniformly dispersed in the TPE matrix than the TPE/3UnTiO₂ system. Changes in both particle size and interface indicate that TPE/3CnTiO₂ system help break down particles and modify interfacial interactions. This result agrees with the outcomes of the tensile features in Figure 1.

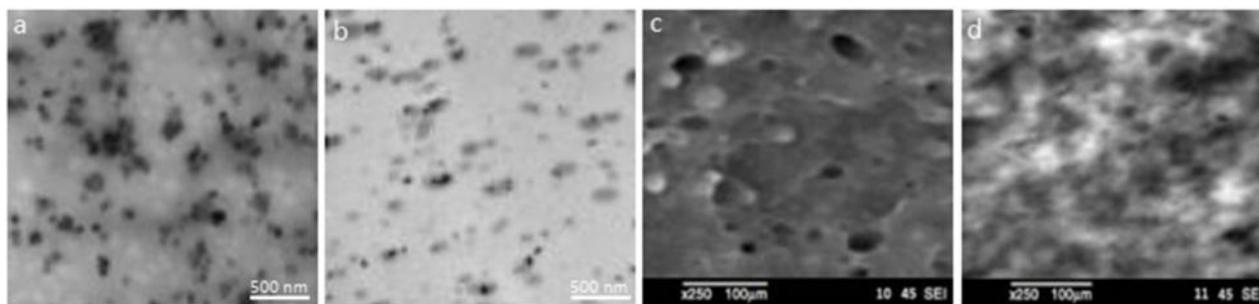


Fig. 2- TEM images of (a) TPE/3UnTiO₂ and (b) TPE/3CnTiO₂ nanocomposites and SEM images of (c) TPE/3UnTiO₂ and (d) TPE/3CnTiO₂ nanocomposites.

3.3. Thermal properties of the nanocomposites

The results of the DSC of the TPE matrix and its nanocomposites are given in Table 1. As it can be noted from Table 1, the crystallization temperature (T_c) has improved from 136.5°C of TPE to 138.3°C of TPE/3UnTiO₂ nanocomposite due to the heterogeneous nucleation of nTiO₂. Significantly, TPE/3CnTiO₂ increased the T_c to 141.2°C, much higher than TPE and even higher TPE/3UnTiO₂ due to the presence of surface carboxylated nTiO₂ favored the crystallization and provided better distribution of the nanoparticles. The interaction between TPE and surface carboxylated nTiO₂ was better than that of UnTiO₂ because during the melting, the acidic functionality of the polymer with -COOH moieties on the surface of the carboxylated nTiO₂ particles interacted with the TPE mode and the nTiO₂ through the covalent bonds. The TPE/3UnTiO₂ nanocomposite with 3CnTiO₂ has an obvious decrease of ΔH_c and ΔH_m . This is due to the formation of the chemical bonds between TPE and nTiO₂, which prevent the movement of the PBT molecules, thus decrease the crystallinity of the nanocomposites. Moreover, the inclusion of TiO₂ nanoparticles in TPE displays a slight effect on the melting temperature (T_m). At 3-wt% of TiO₂ nanoparticles, the T_m increases from 183.7°C of the matrix to 188.9°C of TPE/3UnTiO₂ nanocomposites. The improvement is about 5.2°C. The increase in T_m in nanocomposite should be attributed to the chemical linkage between the matrix introduced in Scheme 1 and the nTiO₂ particle, which increases the chain length and the molecular weight of the TPE. Moreover, the T_m of the TPE matrix was improved by the inclusion of CnTiO₂ seemingly. Take into account the nucleation effect of CnTiO₂ in TPE; one can consider that the crystallization of TPE in TPE/3CnTiO₂ is mostly measured by CnTiO₂ somewhat than by UnTiO₂.

Table 1: Thermal properties of pure TPE and TPE/nTiO₂ nanocomposites.

Nanocomposites (w/w %)	T_m (°C)		T_c (°C)	ΔH_c (J/g)	T_{onset} (°C) (TGA)
	A	B	B		
Pure TPE	90.2	178.3	136.5	28.3	385.6
TPE/3UnTiO ₂	97.6	183.7	138.3	27.2	391.4
TPE/1CnTiO ₂	99.4	186.2	139.6	25.4	394.7
TPE/3CnTiO ₂	104.3	188.9	141.2	24.9	398.5
TPE/5CnTiO ₂	101.2	187.2	140.3	25.1	397.3

A: PTMG block; B: PBT block

The TGA probe provides important evidence in determining the thermal stability of nanocomposites. Thermal stability is a significant property, for which nanocomposite morphology plays an important role (Kilburn et al., 2006). The TGA curves are shown in Figure 3. It realizes that all samples exhibit a one-step decay process in the 330–470°C temperature range. The weight reduction of TPE and its nanocomposite is less than 0.04% with a residual moisture loss of the matrix below 330°C. The key decomposition related to matrix backbone ruin in TPE starts at about 330°C with a hastened rate. The onset decay temperature (T_{onset}) of TPE, which indicates the thermal stability of an ingredient, is 385.4°C. The inclusion of TiO₂ nanoparticles reinforces the thermal stability of the matrix, specifying that the thermal decay temperature of the nanocomposite increase with the nanoparticle's concentration. The maximum increase in thermal stability of TPE/3CnTiO₂ nanocomposite is 12.9°C compared to pure TPE.

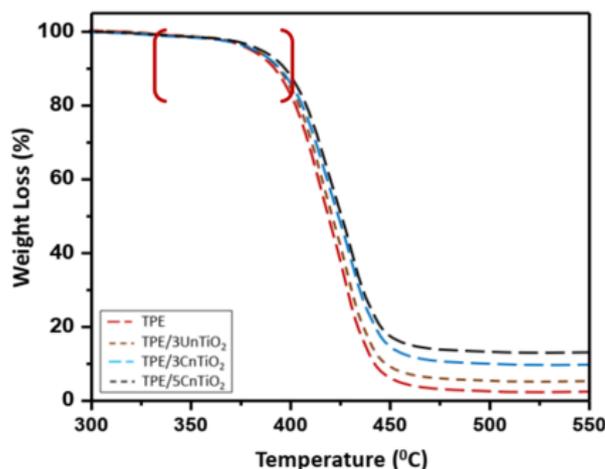


Fig. 3- TGA curves of pure TPE and TPE/nTiO₂ nanocomposites.

4. CONCLUSIONS

The PBT-PTMG block copolymer-based TPE/nTiO₂ nanocomposites were manufactured by a melt mixing process tracked by compression molding. This work examined the influences of unmodified and surface carboxylated nTiO₂ particles on the mechanical, morphological, and thermal properties of TPE/nTiO₂ nanocomposites. The inclusion of nTiO₂ particles in the TPE matrix enhanced the tensile properties of TPE/nTiO₂ nanocomposite compared to TPE. Furthermore, the addition of chloroacetic acid-coated-nTiO₂ particles to TPE significantly improved tensile features such as tensile strength, and impact strength but decreased tensile modulus. When the chloroacetic acid-coated-nTiO₂ particles (CnTiO₂) were at 3 wt%, the tensile strength and impact strength of these nanocomposites reached their maximum values due to better interfacial bond between TPE and nTiO₂ particles. SEM tests showed that 3CnTiO₂ was incorporated into TPE, 3CnTiO₂ particles were evenly distributed in TPE, and their interface bond with TPE was higher than that of TPE/UnTiO₂ in the absence of 3CnTiO₂. DSC measurements showed that the inclusion of 3CnTiO₂ resulted in higher crystallization and melting temperatures enriched simultaneously. The crystallization or melting temperature of TPE/3CnTiO₂ is much greater than that of TPE/UnTiO₂ and that of TPE. The thermal stability of TPE/3CnTiO₂ was higher than that of TPE/UnTiO₂ because there is a compatibility between TPE and CnTiO₂. As a total result of this study, it was sound to confirm that improved morphology, tensile properties and, thermal degradation stability of TPE/nTiO₂ or TPE/CnTiO₂ were due to the existence of compatibility between TPE and nTiO₂ due to a coupling agent.

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